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THE INFLUENCE OF STRAIN RATE ON THE PLASTICITY OF STEEL SHEETS

The paper analyses the influence of strain rate on the behaviour of un-alloyed steels with Re 210–550 MPa in the deformation process. It presents and analyses the results of the influence of the strain rates ranging from 10^{-3} to $2.5 \cdot 10^2 \text{ s}^{-1}$ on the yield point, tensile strength, elongation and reduction of area. It analyses the non-homogeneity of development of plastic deformation from both, the macroscopic and microscopic points of view, as well as the influence of the strain rate on the development of plastic deformation. Since the intensity of the influence of the strain rate on the properties of materials depends on their internal structure, the tested steels are divided into three groups based on their yield point and yield point of tensile strength ratio.

1. Introduction

The strain rate is an external factor, which significantly influences the limit state of a material, its behaviour in the forming process, as well as on the resulting properties of the formed products [1, 2, 3, 4, 11]. The influence of strain rate on the strength characteristics (yield point, tensile strength) are described using parametric equations [2, 5, 12]. However, the sensitiveness of a material on the strain rate is a function of the material structure, and it can generally be stated that the “more homogeneous” structure, the higher its sensitiveness to the strain rate [6, 7]. In general, it applies that increased strain rate results in increased strength characteristics of materials, while the yield point grows more intensively than the tensile strength [6, 13]. As a result, increased strain rate resulted in an increased of R_e/R_m ratio, and for certain materials at critical strain rate is ratio $R_e/R_m \geq 1$, whereas are created conditions of local loss of plastic deformation. This fact significantly influences the formability (in particular pressability) of materials as a result of localization of plastic deformation to “suitable” places.

Plastic deformation is characterized by the fact that its development is markedly non-homogeneous. The non-homogeneity degree is a function of internal and external factors. The internal material microstructure (number and structure of phases, grain size, and structure type) is the internal factor. Increased grain size and number of phases result in significantly increased non-homogeneity of plastic deformation. The temperature, the strain rate and the stress state are crucial external factors [4, 8].

The influence of the strain rate on the development of plastic deformation should be assessed in terms of the deformed body.

From the macroscopic point of view, an increased strain rate results in an increased non-homogeneity of deformation of the body as a whole. From the microscopic and submicroscopic points of view, the situation is opposite [8]. However, it applies that an increased degree of plastic deformation results in an increased degree of non-homogeneity.

Increasing the strain rate in forming semiproducts or products is one of the ways of intensification of production. Therefore, an attention is paid to the study of the influence of the strain rate on the behaviour of material in the deformation process, but also on the methodology of evaluation of formability at increased strain rates (including impact ones) [9, 10].

The sensitivity of materials to the strain rate in the forming process is, as mentioned earlier, a material function, and, therefore, it is useful to analyse this sensitivity especially for new materials intended for cold forming. This is necessary in order to determine the limit state, as well as the properties of the final product. The aim of the paper is to extend the pieces of knowledge and to point out certain problems occurring during forming at increased rates (up to impact loads).

2. Experimental material and procedure

Experimental program related to influence of strain rate on plastic characteristics of steel sheets were realized on steel sheets produced by cold forming process. Steel sheets marked as E280G and H340LAD were cold rolled at thickness of 1 mm and next were hot rolled at the final thickness of 8 mm. Microstructure of tested steels is ferrite-pearlite. From steel sheets were taken samples

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in rolling direction which were for used for tensile test. The main characteristics of tested steels (carbon content, grain size d , and microalloying elements) are shown in Tab. 1

The basic criteria for assessment of the formability of materials obtained during the tensile test including the yield point R_e , the tensile strength R_m , the elongation A , the R_e/R_m ratio.

The presented test result were obtained on the INSTRON tensile testing machine with the loading rate up to 1000 mm/min, as well as on an instrumented pendulum impact testing machine with the loading rate of up to approx. 5 m/s and on a drop tester. By a suitable combination of the loading rate (v) and the deformed length of test bars (L_0), strain rates $\dot{\epsilon} = v/L_0$ ranging between 10^{-4} and 10^3 s^{-1} could be obtained.

The influence of the strain rate on the development of plastic deformation was observed from the macroscopic and microscopic points of view. The influence of the strain rate on the development of deformation was observed on test bars, whose measured length was divided into 1 mm divisions. Such prepared test bars were gradually loaded to a certain strain degree (from approx. 5 to 30%). After each strain degree, the distance between individual divisions was measured. The test bars were loaded statically (10^{-3} s^{-1}) and dynamically (10^2 s^{-1}). The mean non-homogeneity was evaluated,

$$M = \frac{\sum_{i=1}^n U_x \cdot m}{n} \quad \text{where } n \text{ is the number of elements of}$$

the statistic population. The other quantities result from Fig. 8. For evaluation of micro non-homogeneity of plastic deformation, a similar procedure as for evaluation of macro non-homogeneity was used. A metallographic sample was prepared on the test bar stem, and a grid with the mesh size of $0.05 \times 0.05 \text{ mm}$ was created using a microhardness tester in the area of $1 \times 0.5 \text{ mm}$. The mean strain ϵ was measured on the length of 1 mm, and also on each division at individual strain degrees from 5 to 35%.

3. Influence of the strain rate on the basic mechanical properties

The influence of the strain rate on the basic mechanical properties of the tested steel C4 is shown in Fig. 1, which indicates that increased strain rate results in increased strength properties, while the intensity of growth of R_e is higher than that of R_m . The dependence of the strength properties on the strain rate for unalloyed high-grade steels in the interval from 10^{-3} to 10^3 s^{-1} was described using parametric equations (1), (2), mostly in the shape presented in [2, 5, 7],

$$R_{e\dot{\epsilon}} = R_{e\dot{\epsilon}_0} + A \cdot \ln(\dot{\epsilon}/\dot{\epsilon}_0) \quad (1)$$

$$R_{m\dot{\epsilon}} = R_{m\dot{\epsilon}_0} + B \cdot \ln(\dot{\epsilon}/\dot{\epsilon}_0) \quad (2)$$

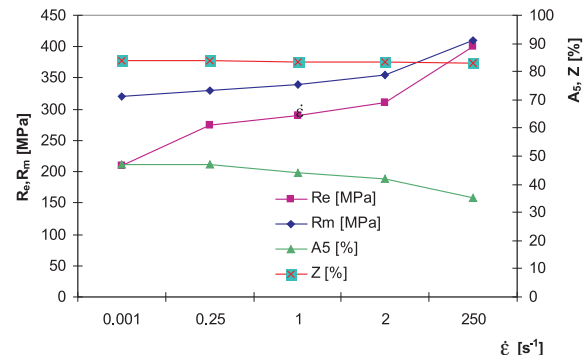


Fig. 1 Influence of the strain rate $\dot{\epsilon}$ on mechanical properties of steel C4

where R_e , R_m are the yield point and the tensile strength at the respective strain rate $\dot{\epsilon}$, and $R_{e\dot{\epsilon}_0}$, $R_{m\dot{\epsilon}_0}$ are the yield point and the tensile strength at the static strain rate (10^{-3} s^{-1}). The A and B parameters are material constants characterizing the sensitivity of the material to the strain rate. Fig. 2 shows the influence of the strain rate on the increment of the yield point ΔR_e of steel C33 after various heat treatments. The as-quenched steel has the lowest sensitivity to $\dot{\epsilon}$, because the martensitic structure has the greatest number of obstructions to dislocation movement, and the as-nor-

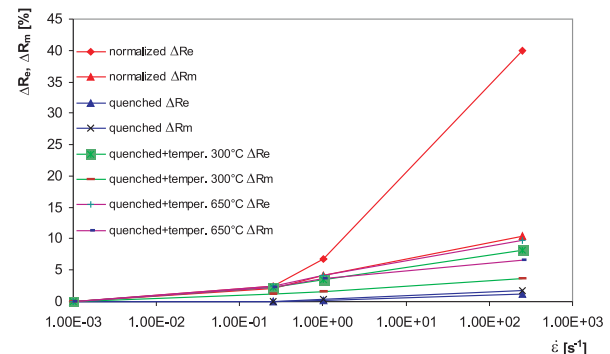


Fig. 2 Influence of the strain rate $\dot{\epsilon}$ on the increment of strength properties ΔR_e , or ΔR_m , after heat treatment, compared with the initial state ($\dot{\epsilon} = 10^{-3} \text{ s}^{-1}$) of steel C33, after various heat treatments

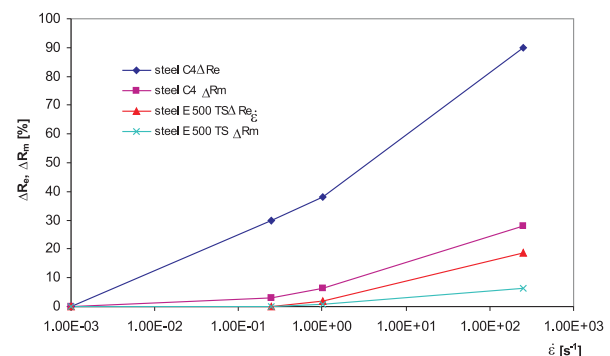


Fig. 3 Influence of the strain rate $\dot{\epsilon}$ on the increment of strength properties ΔR_e , or ΔR_m , compared with the initial state at $\dot{\epsilon} = 10^{-3} \text{ s}^{-1}$ for various steel grades.

Mechanical properties of tested steels at characteristic strain rates.

Table 1.

Tested steel	Characteristics	$\dot{\epsilon} \text{ s}^{-1}$			
		10^{-3}	$2.5 \cdot 10^{-1}$	1	$2.5 \cdot 10^2$
C4 C<0.04% d = 0.031 mm	R_e [MPa]	210	275	290	400
	R_e/R_m	0.65	0.8	0.83	0.93
	A_5 [%]	47	46	44	35
E280G C<0.04% d = 0.009 mm	R_e [MPa]	295	340	350	459
	R_e/R_m	0.70	0.79	0.80	0.9
	A_5 [%]	30	27	25	22
H340LAD Nb,V<0.1% d = 0.008 mm	R_e [MPa]	350	380	430	540
	R_e/R_m	0.81	0.79	0.82	0.94
	A_5 [%]	26	26	26	24
C10 C<0.10% d = 0.022 mm	R_e [MPa]	280	340	360	490
	R_e/R_m	0.63	0.74	0.78	0.94
	A_5 [%]	41	40	40	40
C28 C<0.28% d = 0.015 mm	R_e [MPa]	370	410	420	540
	R_e/R_m	0.68	0.74	0.75	0.90
	A_5 [%]	30	30	31	33
C33 C<0.33% d = 0.012 mm	R_e [MPa]	440	451	470	630
	R_e/R_m	0.62	0.62	0.63	0.80
	A_5 [%]	27	26	26	27
C33 quenched	R_e [MPa]	1910	1910	1910	1935
	R_e/R_m	0.81	0.81	0.83	0.81
C33 quenched +temp. 300°C	R_e [MPa]	1480	1525	1540	1600
	R_e/R_m	0.92	0.95	0.95	1.0
C33 quenched +temp. 550°C	R_e [MPa]	820	840	845	900
	R_e/R_m	0.90	0.90	0.90	0.90
X350M C<0.1% Nb,V d = 0.010 mm	R_e [MPa]	380	420	440	590
	R_e/R_m	0.79	0.80	0.83	0.1
	A_5 [%]	34	34	31	30
X420M C<0.1% Nb,V d = 0.0062 mm	R_e [MPa]	430	485	500	610
	R_e/R_m	0.75	0.79	0.8	0.87
	A_5 [%]	30	30	30	30
X500M C<0.1% Nb,V d = 0.0037 mm	R_e [MPa]	550	550	550	640
	R_e/R_m	0.86	0.86	0.86	0.88
	A_5 [%]	28	28	28	28
X560M C<0.11% Nb,V d = 0.009 mm	R_e [MPa]	570	570	570	650
	R_e/R_m	0.87	0.87	0.87	0.89
	A_5 [%]	28	28	28	28
S315MC C<0.05% Nb,V d = 0.007 mm	R_e [MPa]	390	420	440	570
	R_e/R_m	0.82	0.82	0.82	0.9
	A_5 [%]	38	38	38	37
S460MC C<0.07% Nb d = 0.006 mm	R_e [MPa]	537	550	592	700
	R_e/R_m	0.86	0.86	0.86	0.87
	A_5 [%]	30	30	30	29
E500TS C<0.08% Nb,Ti d = 0.0056 mm	R_e [MPa]	540	540	550	640
	R_e/R_m	0.84	0.84	0.85	0.94
	A_5 [%]	39	39	39.5	40

malized steel has the highest sensitivity. Similarly, Fig. 3 documents the influence of $\dot{\epsilon}$ on ΔR_e and ΔR_m of steels C4 and E500TS with different grain sizes. The grain boundaries are insuperable obstructions to dislocation movement, therefore the finer grain the more obstructions, and the steel is less sensitive to the strain rate $\dot{\epsilon}$.

tions to dislocation movement, therefore the finer grain the more obstructions, and the steel is less sensitive to the strain rate $\dot{\epsilon}$.

In terms of assessment of formability, the R_e/R_m ratio is the most important criterion. Table 1 shows ratio R_e/R_m of the tested steels at four testing conditions (static 10^{-3} s^{-1} , two quasi-static $2.5 \cdot 10^{-1}$ and 1 s^{-1} and dynamic $2.5 \cdot 10^2 \text{ s}^{-1}$) strain rates.

Based on the R_e/R_m ratio at the static tensile test, the tested steels can be divided into three groups: steels with $R_e < 300 \text{ MPa}$ and $R_e/R_m < 0.7$, steels with $R_e > 300 \text{ MPa}$ and $R_e/R_m > 0.7$, and the third group is formed by steels with $R_e > 500 \text{ MPa}$ and $R_e/R_m > 0.8$.

The tests results shown in Tab.1, obtained on various steel grades, confirm that an increased strain rate results in an increased R_e/R_m ratio and the intensity of this increase is a material function. The fewer obstructions to dislocation movement in a material, the more marked intensity of the influence of $\dot{\epsilon}$. Fig. 4 shows an example how the grain size influences the R_e/R_m ratio at various strain rates.

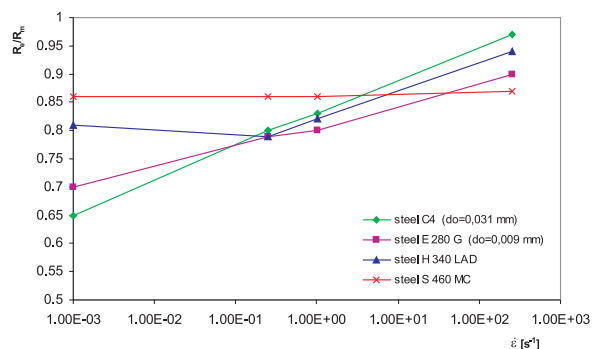


Fig. 4 Influence of the strain rate $\dot{\epsilon}$ on R_e/R_m for various steel grades.

The analysis of the results of strength properties (R_e , R_m) obtained from tensile test shows that tested steels in strain rates ranging of 10^{-3} to $2.5 \cdot 10^2 \text{ s}^{-1}$ shows the R_e/R_m ratio < 1 and in terms of the macro-volume, up to this strain rate the plastic stability can be maintained up to the strain value corresponding to R_m . The critical value $\dot{\epsilon}_{cr}$ at the ratio R_e/R_m will equal 1, is influenced by the internal material structure and it can generally be stated that the higher R_e , which is considered to be a macroscopic characteristic of the steel structure, the higher $\dot{\epsilon}_{cr}$. As proven by the tests made on high-grade unalloyed steels with R_e from 210 to 350 MPa (E280G, H340LAD), $\dot{\epsilon}_{cr}$ is in the interval from 10^2 to $3 \cdot 10^2 \text{ s}^{-1}$.

4. Influence of the strain rate on the homogeneity of plastic deformation

The elongation (A) and the reduction of area (Z) are the macroscopic characteristics of the plasticity of steel. Fig. 5 shows the graphic relationships $\dot{\epsilon} - A$ and $\dot{\epsilon} - Z$. Tab. 1 shows the values of these quantities at the characteristic strain rates. In terms of the

influence of the strain rate in the interval from 10^{-3} to $2.5 \cdot 10^2 \text{ s}^{-1}$ on the above-mentioned characteristics, the tested steels can be classified under the group of steels where the elongation decreases with the growing strain rate, and under the group where the elon-

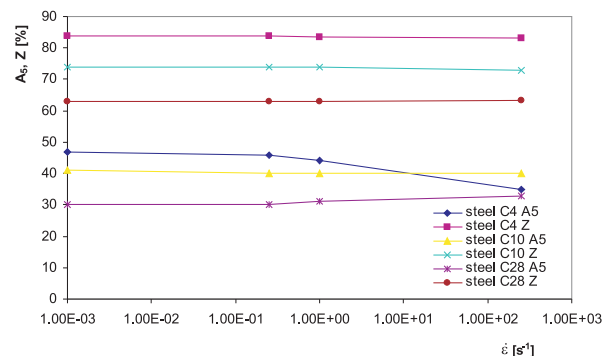


Fig. 5 Influence of the strain rate $\dot{\epsilon}$ on the elongation A_5 and the reduction of area Z of various tested steel grades

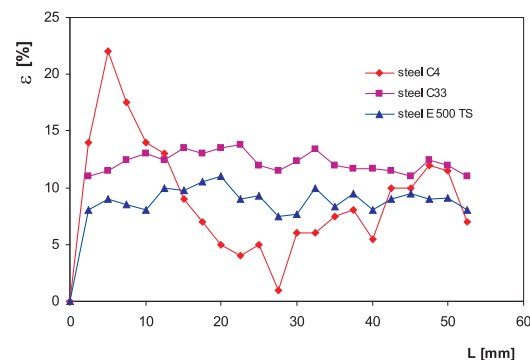


Fig. 6 Distribution of plastic deformation along the test bar at impact loading 10^2 s^{-1}

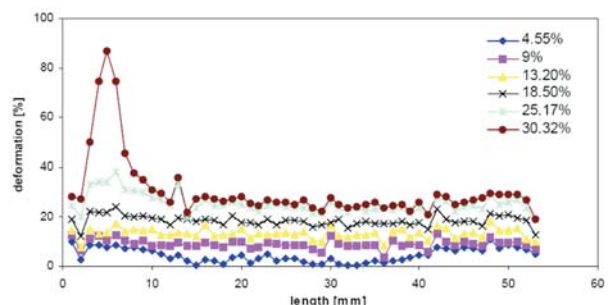


Fig. 7 Distribution of deformation along the test bar at various total strain degrees and impact loading 10^2 s^{-1} of steel 315MC

gation does not change, or even grows. A decrease in the elongation with a growing $\dot{\epsilon}$ is only shown by steels with $R_e < 300 \text{ MPa}$ after exceeding $R_e/R_m > 0.82$. Steels with a higher yield point

maintain or even increase their elongation at $R_e/R_m > 0.82$, too. The non-homogeneity of distribution of deformation along the test bar is influenced, besides the R_e/R_m ratio, also by the $R_m - R_e$ difference. At a little difference, a very non-homogeneous distribution of deformation along the test bar, and hence a decreased elongation, can take place. This assumption is confirmed by Fig. 6, which shows the distribution of deformation along the test bar at the strain rate of 10^2 s^{-1} for steel C4, for which $R_m - R_e = 10 \text{ MPa}$, steel C33, for which this difference is 160 MPa, and for steel E500TS, where this difference is 100 MPa. Fig. 7 documents the results of measuring the strain distribution along the test bars made of steel S315MC at individual degrees of total strain and dynamic loading. The results of the influence of the strain rate on the criteria of macro non-homogeneity M of the tested steels are presented in Fig. 9. The results show that the non-homogeneous strain distribution along the test bar increases with an increased strain degree. The influence of the strain rate on the strain distribution along the test bar depends on the properties of steel. If R_e/R_m approaches to 1 at a given strain rate, then the non-homo-

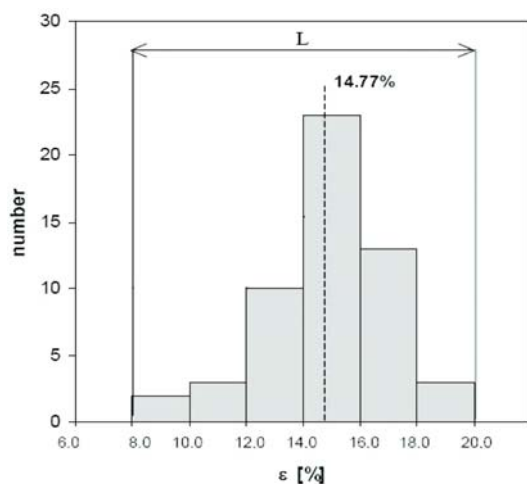


Fig. 8 Histogram of strain distribution at the mean strain $\varepsilon = 14.77\%$ of steel 315MC at the strain rate of 10^{-3} s^{-1}

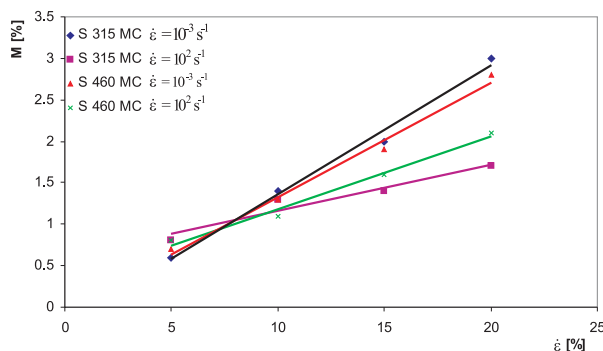


Fig. 9 Relationship between the mean macro non-homogeneity M and the mean strain degree in a micro-area ($1 \times 0.5 \text{ mm}$) ε at the strain rates of 10^{-3} and 10^2 s^{-1} of selected steels

geneity of strain distribution along the test bar significantly increases (see Fig. 6, steel C4). If $R_e/R_m < 1$ (0.82 to 0.9), the strain rate favourably influences the strain distribution along the test bar (Fig. 6 and Fig. 9).

The results of the influence of the strain rate on the criteria for micro non-homogeneity of M and L are presented in Fig. 10 and Fig. 11. L is the range of the statistic population (Fig. 8). The results confirms that the micro non-homogeneity increases with an increasing strain degree.

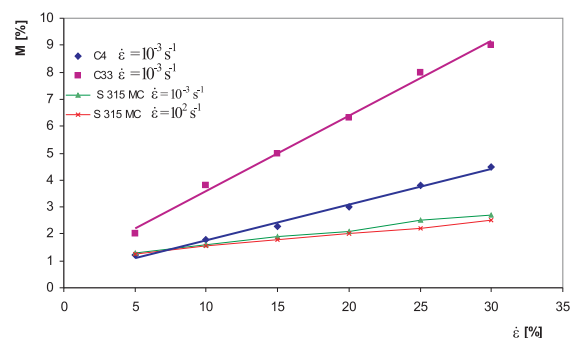


Fig. 10 Relationship between the mean micro non-homogeneity M and the mean strain degree in a micro-area ($1 \times 0.5 \text{ mm}$) ε at the strain rates of 10^{-3} and 10^2 s^{-1} of selected steels

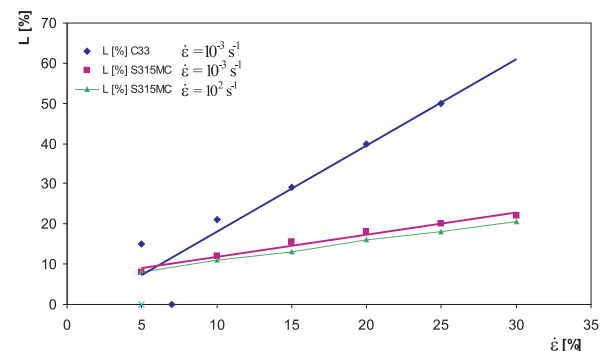


Fig. 11 Relationship between the range of the statistic population L and the mean strain degree in a micro-area ε at the strain rates of 10^{-3} and 10^2 s^{-1}

The results of evaluation of non-homogeneity of plastic deformation show that in the interval of these strain rates from 10^{-3} to 10^2 s^{-1} the non-homogeneity slightly decreases or does not change with an increasing strain rate of the tested steels. This means that forming these steels at higher rates practically does not influence the properties of these products.

The intensity of increase in the non-homogeneity of plastic deformation with an increasing strain degree is a function of the structure, as shown by the results. The degree of non-homogeneity of the steel C33 with the ferrite and pearlite structure is as many as three-times higher than that of steel C4 with the ferrite

structure. It results from Fig. 9–11 that these relationships can be described using the formulae $M = a + k_1 \cdot \varepsilon$, $L = b + k_2 \cdot \varepsilon$ (k_1 and k_2 are material constants). Here it also applies that the less obstructions to dislocation movement in steel, the more non-homogeneous its deformation process, but this deformation process is practically not influenced by the increase in the strain rate.

5. Conclusions

The aim of the paper was to assess the influence of the strain rate in the interval from 10^{-3} to $2.5 \cdot 10^2 \text{ s}^{-1}$ on the mechanical properties, with regard to the plasticity of un-alloyed high-grade steels with the yield point from 210 to 550 MPa. Based on the analysis of experimental results obtained from a long time and literature knowledge, the following can be stated:

- The resistance of materials to plastic deformation increases with an increasing strain rate, which increases the strength properties of the tested steels and the yield point to tensile strength ratio R_e/R_m .

- The intensity of increase in R_e/R_m with an increasing strain rate is a function of the internal material microstructure. R_e . The intensity of increase in R_e/R_m with an increasing strain rate is the highest at steels with $R_e < 300 \text{ MPa}$, lower at steels with $R_e < 500 \text{ MPa}$, and slight at steels with $R_e > 500 \text{ MPa}$
- The influence of the strain rate on the plasticity (elongation and reduction of area) values is based on R_e/R_m . Only steels with $R_e < 300 \text{ MPa}$ show a decreased elongation in the observed strain rate interval, from the strain rate where $R_e/R_m > 0.82$. Steels with a higher yield point maintain or increase their elongation.
- The non-homogeneity of plastic deformation increases with an increasing strain degree. The intensity of macro and micro non-homogeneity is a function of the internal material microstructure. An increasing strain rate slightly decreases the micro non-homogeneity of plastic deformation.

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